

SCMT5

Kingston University London, UK, July 14-17, 2019

SULFATE RESISTANCE OF PORTLAND CEMENT MORTARS: A COMPARISON OF NANO AND MICRO SILICA

Nader Ghafouri Ph. D.¹, Iani Batilov, P.E.², Meysam Najimi, Ph. D.³

¹*Professor, Dep. of Civil and Environmental Engineering and Construction, Univ. of Nevada Las Vegas, NV 89154, USA*

²*Civil Engineer, Stantec Consulting Services, Las Vegas, NV 89102, USA*

³*Postdoctoral Research Associate, Dep. of Civil, Construction and Environmental Engineering, Iowa State University, IA 50010, USA*

ABSTRACT

Presented is a direct comparison sulfate resistance study of mortars, containing 3 or 6% cement replacement with either colloidal nanosilica (nS) or microsilica (mS), exposed to 26 weeks of full submersion in a 5% sodium sulfate (Na₂SO₄) solution. Mortar bar samples prepared per ASTM C 490 and measured for linear expansion per ASTM C 1012 over the 6-month testing period indicated that at 6% cement replacement, colloidal nS exhibited on average 75% the expansion of the microsilica containing counterpart. At 3% replacement, either form of silica reduced the sulfate attack related expansion to a similar degree (by 35±2%) in comparison to the control mixture. Supplemental rapid sulfate permeability testing (RSPT) supported the expansion results. The 6% nS mortar mixture exhibited the least charge passed which indicated that the nS contained mortar were more impermeable in nature and more resistant to ion transport. SEM images taken of the nS and mS particles at similar levels of magnification visually revealed the significant difference in particle size between both forms of silica. Absorption testing per ASTM C 642 revealed a larger permeable pore volume in the 6% mS containing mortar mixture in comparison to 6% nS. The smaller permeable pore volume of the 6% nS mixture supports the physical paste densification effect observed with nS application in cementitious mixtures from other studies. Mercury intrusion porosimetry (MIP) tests of the control and 6% silica containing mixtures, revealed evidence of paste and paste-to-aggregate interfacial zone densification, as well as pore size refinement. Through the use of nS and mS, the void volume proportion of gel pores and capillary micropores in the silica containing mixtures increased in comparison to the control, and there was an overall reduction in the capillary macropores.

Keywords: nanosilica, microsilica, silica fume, pozzolan, durability, sulfate attack, sulfate resistance

INTRODUCTION & BACKGROUND

As succinctly yet effectively stated in a book on durable concrete, “structural failures are rare, but durability failures are all too common (Richardson 2002).” That is often the case when due consideration is not given to specifying concrete mixture designs well-suited for the environment under which the structure will be subjected to and achieving a long and ideally maintenance-free service life. In the recent years it has been well documented that cement production is the third-largest source of human generated CO₂, and the global cement production has only accelerated having grown to more than 30-times that reported in 1950 (Andrew 2018). There is an economic and sustainable incentive to build durable structures using concrete mixture designs well-suited for the anticipated effects of weathering, abrasion, and exposure to the chemical and physical assault from ASR, chlorides, or sulfates. The cost of repairing, protecting, and rehabilitating concrete infrastructure in the US alone is estimated to be upwards of \$21 billion (ICRI 2006). The ability to utilize the existing concrete structures up to and beyond their intended service life and allow for cost-effective retrofitting that does not require complete demolition and expenditure of the energy and resources required to demolish, dispose, and rebuild is now a major focus of research and development (Richardson 2002).

Silica fume or microsilica (mS) as referred to from here out, is one of the well-known industry recognized supplementary cementitious materials (SCMs) used in the production of high-durability concrete (Kosmatka and Wilson 2016). Numerous research has been published since the pozzolanic byproduct, condensed from the SiO₂ rich vapors of electric arc furnaces used in the silicon and ferrosilicon industry, was first collected and studied in Norway in the late 1940s and later brought to the US in the 1980s (ACI Committee 234 2006). Currently, mS is a code recognized pozzolan recommended by ACI 318 for concrete anticipated to come in contact with water or soil containing deleterious concentrations of sulfate that meet the S3 sulfate exposure classification (ACI Committee 318 2014). Nanosilica (nS) is among the collection of new engineered nanoscale materials investigated for use in cementitious materials and shown to improve concrete strength and durability properties (Du et al. 2014; Ghafoori et al. 2016; Sanchez and Sobolev 2010). Chemically it is very similar to mS, as it is essentially nano-sized (<100nm) silicon dioxide (SiO₂) particles. In comparison, mS particles are larger, typically < 1 μm (Holland 2005). The smaller particle sizes of nS correlate with a specific surface area of 80 m²/g or more while that of mS is typically in the 15-25 m²/g range (Campillo et al. 2004). The higher surface area makes nS a much more reactive pozzolan that can rapidly and effectively limit available Ca(OH)₂ for reaction with sulfates. Furthermore, nS has been shown to create seeding sites for the alite (C₃S) and belite (C₂S) phase during hydration that contributes to the growth of a more compact C-S-H phase and a densified cement paste to aggregate interfacial zone (Sanchez and Sobolev 2010; Singh et al. 2013).

External sources of either sodium-, potassium-, magnesium-, and calcium sulfate are common in the soil, groundwater, or seawater the concrete will be in contact with (Skalny et al. 2002). The effects of sulfate attack are expansion, spalling, softening, adhesion loss, and decalcification of the hydrated phases (Mehta 2000; Skalny et al. 2002; Wee et al. 2000). Sulfate attack could compromise the protective concrete cover and increase permeability which will result in more significant issues once a path to the reinforcement steel is established and corrosion begins. It was the objective of the researchers to evaluate if nS would offer a

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comparable or superior performance to mS when specifying pozzolan for durable concrete mixture designs intended for structures subject to severe sulfate exposure conditions.

EXPERIMENTAL PROGRAM

Materials

The Portland cement used was locally sourced and Type II per ASTM C 150 (ASTM International 2002a) with an inherently moderate sulfate resistance due to the limited C₃A content of 7.2%. The mS was sourced from a US supplier in an undensified gray amorphous sub-micron powder form. The pozzolan met all chemical and physical requirements per ASTM C 1240 (ASTM International 2003a). The nS used was in the form of a commercially available aqueous dispersion containing 25% by weight 5-35 nm (0.197-1.378×10⁻⁶ in) amorphous nS particles. The chemical and physical properties of the cement, microsilica, and nanosilica are presented in *Table 1*.

The fine aggregate used for the mortars in this study was from a Nevada based quarry and had an oven-dry specific gravity of 2.76 as measured by ASTM C 128 (ASTM International 2015), absorption of 0.81% and a fineness modulus of 2.64. Its gradation was well inside the upper and lower limits of ASTM C 33 (ASTM International 2003b). Commercially bottled distilled water purchased from a single source was used for mortar mixing and preparation of the 5% sodium sulfate solution. A polycarboxylate based high-range water-reducing admixture (HRWRA) was utilized for achieving the desired flow per ASTM C 109 (ASTM International 2002b).

Table 1: Properties of Cementitious Materials

	Type II Cement	Microsilica (mS)	Colloidal Nanosilica (AQnS)
<i>Chemical Composition</i>			
Silicon Dioxide (SiO ₂), %	21.1	94.72	99.9
Aluminum Oxide (Al ₂ O ₃), %	4	--	--
Ferric Oxide (Fe ₂ O ₃), %	2	--	--
Calcium Oxide (CaO), %	62.7	--	--
Magnesium Oxide (MgO), %	2.1	--	--
Sulfur Trioxide (SO ₃), %	2.8	0.23	--
Loss on Ignition, %	1.8	2.82	--
Insoluble Residue, %	0.71	--	--
Total Alkali (Na ₂ O + K ₂ O), %	0.59	0.49	--
Free Lime (CaO), %	0		
<i>Physical Properties</i>			
Time of Set Initial Vicat, min	145	--	--
Specific Surface Area, m ² /g	0.341 ^a	22.65 ^b	--
325 Mesh (45 μm), % passing	--	97.12	
Avg. Particle Size (APS), μm	20-30 ^c	0.1-1.0 ^c	0.005- 0.035
<i>Per Bogue Calculation^d</i>			
Tricalcium Silicate (C ₃ S), %	57.0	--	--

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Dicalcium Silicate (C ₂ S), %	17.5	--	--
Tricalcium Aluminate (C ₃ A), %	7.2	--	--
Tetracalcium Aluminoferrite (C ₄ AF), %	6.1	--	--

^aby Blaine air-permeability test^bby BET Analysis^cEstimated from MasterSizer Particle Distribution Analysis^dBogue Modified Equation for Interground Gypsum & Limestone (Winter 2012)1 μm = 0.03937×10⁻³ in 1 m²/g= 4879ft²/lb**Mixture Design**

A total of five mortar mixtures were tested within the scope of this study, one control mixture with no pozzolan content and two mixtures with either 3% or 6% cement replacement with mS or nS. The mixture proportions are presented in *Table 2*. The water-to-binder ratio (w/b) was kept at a constant 0.485 for all mixtures according to ASTM C 1012 (ASTM International 2004). The fine aggregate-to-binder ratio was 2.75-to-1 by mass as specified in ASTM C 109 (ASTM International 2002b).

Table 2: Mortar Mixture Proportions

Mixture Designation	Binder Content, %			Measured Flow, %*	HRWRA Used, mL	3-Day Compressive Strength,	
	Cement	nS	mS			psi	MPa
CNTL	100	--	--	148%	0.0	4,296	29.6
3mS	97	--	3	108%	0.0	4,420	30.5
6mS	94	--	6	95%	4.0	4,463	30.8
3AQnS	97	3	--	80%	13.0	6,290	43.4
6AQnS	94	6	--	49%	30.0	6,473	44.6

*Flow measured according to ASTM C 1437 with flow table conforming to ASTM C 230

Mixing Procedure and Curing

The mixtures were prepared in an epicyclic mechanical mixer following the mortar preparation procedure of ASTM C 305 (ASTM International 1999). Before combining the binder with water and starting the mortar mixing regimen, the mS was homogeneously intermixed with the cement by hand, or in the case of the colloidal nS, combined with the mixing water before adding the cement to the mixer bowl. For each of the design mixtures, four mortar expansion bars were prepared per ASTM C 1012, 5 cm (2-in) mortar cubes specimens were prepared per ASTM C 109 for strength testing, and six 10 cm (4-in), diameter disks for supplemental testing. For the nS and mS replacement mixtures, addition of HRWRA was required to achieve workability as close to the ASTM C 109 recommendation as possible. HRWRA dosages and flow measurements are

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presented in *Table 2*. All mortar sample molds were manually filled and compacted using an electromagnetic vibrating table. The sample molds for each mortar mixture were then plastic wrapped and kept at room temperature (21 ± 3 °C) for 24 hours. Curing was continued for 3 days in a moist room to achieve the required compressive strength of 20 ± 1.0 MPa (2900 ± 145 psi) per ASTM C 1012 prior to sulfate exposure. Following the 3-day curing period, the mortar bars were transferred to a 5% sodium sulfate solution tank. Disks were kept in the moist curing room prior to using them for the absorption and RSPT tests.

Sulfate Solution

The 5% sodium sulfate solution was prepared per the outlined procedure in ASTM C 1012 (ASTM International 2004). The tank size selected and volume of solution prepared were sufficient to maintain all samples fully submerged and meet the recommended minimum solution to mortar volume ratio of 4. The solution in each container was kept in circulation using submersible pumps and the pH was continuously rebalanced to maintain 6.5 ± 1 using a pH controller and peristaltic pump system that dosed 0.5N H₂SO₄ as needed during the 6 month fully submerged test. The latter measure maintains solution acidity and replenishes the available sulfate ion concentration as established in prior existing studies (Mehta 1975).

Expansion

The four mortar bars prepared for each mixture design were measured in a length comparator with a digital gage following the ASTM C 1012 frequency and measurement procedure. When measured, the mortar bars for each mixture were kept immersed in a portable container filled with sulfate solution from the main test tank to avoid any effects from drying and shrinkage. The reported length change values were calculated as directed in ASTM C 490 (ASTM International 2000) and based on the average between the four bars at each age.

Absorption

Three of the mortar disk specimens prepared were used for the water absorption test after 28 days of curing. The test procedure and calculation of the volume of permeable pore space percentage followed ASTM C 642 (ASTM International 1997). The values presented in the results are based on an average.

Rapid Sulfate Permeability Test (RSPT)

The RSPT is a performance-based six-hour test recommended by the Cement Concrete & Aggregates Australia in a technical note on developing performance-based specifications for sulfate-resisting concrete (CCAA 2011) originally proposed by Tumidajski and Turc (Tumidajski and Turc 1995). The test method is analogous to the rapid chloride permeability test (RCPT) per ASTM C 1202 (International 1997). Instead of 3% NaCl solution on one face of the specimen across the 0.3N NaOH solution, the RSPT test utilizes a 10% Na₂SO₄ solution. The test was performed on mortar disks at the same 28-day curing age as done so for absorption.

Mercury Intrusion Porosimetry (MIP)

Mortar fragments no larger than 15 cm³ were collected from the center of samples to avoid any mold effects, and then oven dried and vacuum desiccated to prepare them for the MIP test. The samples were collected from cubes that had cured the full 6 months and were not exposed to sulfate solution so that the precipitation of sulfate attack related compounds do not factor into the porosity and pore size distribution of the tested mixtures.

Field Emission Scanning Electron Microscopy (FESEM)

For the images of the nano- and microsilica, a model JSM-6700F Field Emission Scanning Electron Microscope (FESEM) with a magnification range of x500 to x430,000 (5 μm to 10 nm) was used (see *Figure 1*). Small samples of the pozzolan were then gold coated with a thin, approximately 20 nm, layer of gold for conductance using an automated sequence sputter coating machine. The nS sample was the collected residue remaining after a few drops of colloidal nS were left to evaporate.



Figure 1: FESEM Used for Imaging of mS and Colloidal nS

RESULTS AND DISCUSSION

The expansion readings recorded for each mortar mixture are presented in *Figure 2* and summarized in *Table 3* for select period of exposure that will be referenced in the discussion. By the fourth week of exposure in the sulfate solution, the pozzolan containing mixtures discernably exhibited less expansion in comparison to the cement only CNTL mixture. At four weeks exposure, the 3mS and 6mS mixtures exhibited 77.3% and 90.9% of the expansion observed with the CNTL, respectively. The AQ3nS and AQ6nS mixtures even more so, showing only 69.7% and 48.5% the expansion compared to the CNTL. Lower levels of expansion correlate to a higher resistance to external sulfate attack. With Na_2SO_4 as the sulfate source, the expansion due to sulfate attack is the effect of excessive gypsum and ettringite formation, both products of the chemical reactions between sulfate ions (SO_4^{2-}) and hydrated cement phases (Hewlett and Massazza 2003; Skalny et al. 2002).

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Table 3: Expansion Measurements at Select Periods of Exposure

Duration of Exposure	Expansion, %				
	CNTL	3mS	6mS	AQ3nS	AQ6nS
4 weeks	0.011	0.009	0.010	0.008	0.005
8 weeks	0.016	0.012	0.015	0.014	0.011
12 weeks (3 months)	0.021	0.017	0.017	0.015	0.013
26 weeks (6 months)	0.039	0.028	0.028	0.027	0.023

*Expansion values are rounded to the nearest thousandth.

Over the duration of the experiment, the mS and nS containing mortar bars continued to exhibit less expansion over the CNTL, yet after 3 months of exposure it became apparent that the expansion of the 3mS and 6mS mortars was very comparable. The percent difference between the two over the first 8 weeks averaged 19%, that difference dropped to a mere 3% by then end of the test period. Doubling the mS replacement content did not result in any improvements in sulfate resistance. On the other hand, the increase of the colloidal nS content from 3% to 6% had a positive effect on the sulfate resistance for that mortar mixture when compared to AQ3nS. Consistently, AQ6nS exhibited less expansion than AQ3nS, on average approximately 80% that of AQ3nS. At 6 months, AQ6nS exhibited an expansion of 0.023% and AQ3nS showed 0.027%, that is a 16% difference.

Comparing the two pozzolans, at 3% replacement, mS and AQnS showed very similar levels of expansion as evident in *Figure 2*. The expansion measurements of mortar mixture 3mS and AQ3nS were 71.8% and 69.2% that of the CNTL, respectively, a mere 4% difference. As stated earlier, at the larger 6% replacement, the mS mixture performed very similarly to the 3% mS, and therefore was surpassed by AQ6nS. The 6% colloidal nS mixture exhibited an expansion of 0.023%, that was 59% the expansion of the CNTL at 6 months and the largest improvement in sulfate resistance. In comparison, 6mS exhibited 71.8% the expansion of CNTL which was almost identical to that of 3mS.

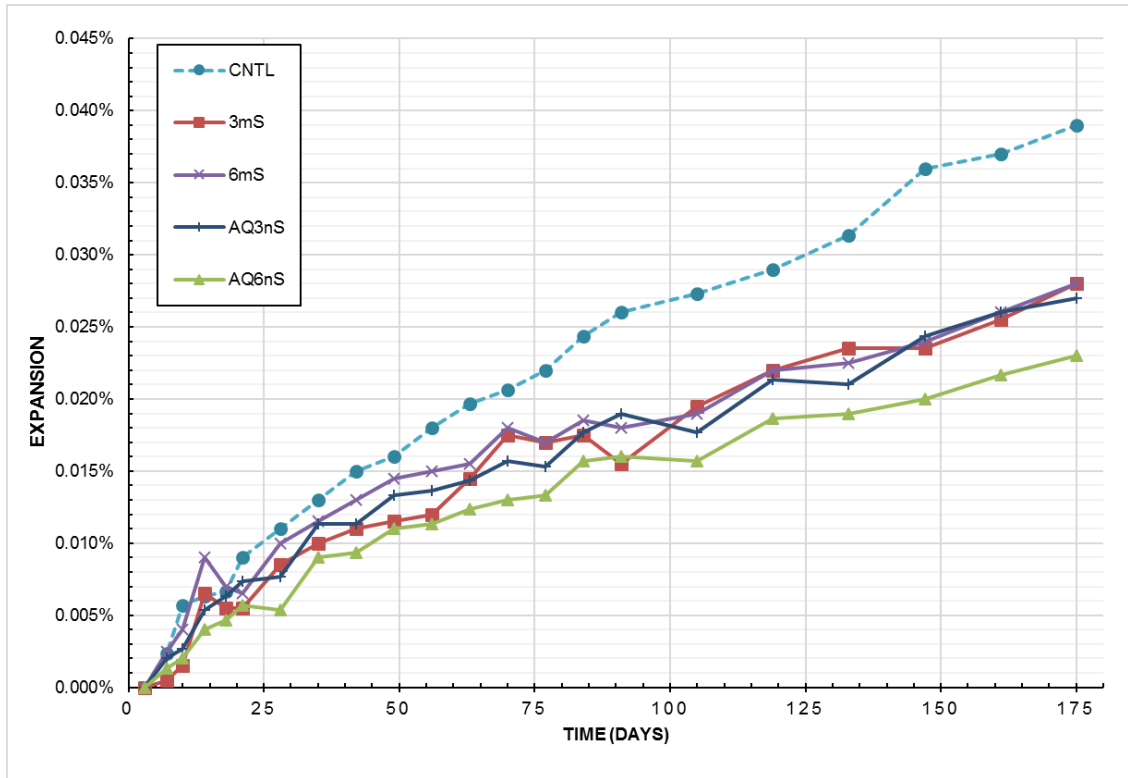


Figure 2: Expansion Measurements Over 6-Month Testing Period

The expansion results were supported by the RSPT test which reflects both the interconnectivity of the mixture's pore structure and free ion movement (Stanish et al. 1997; Tumidajski and Turc 1995). The electrochemical conductance test results, reported as total charge passed in Coulombs, reflects the permeability and diffusivity of all ions through the mortar matrix which includes free sulfate (SO_4^{2-}), hydroxide (OH^-), and calcium (Ca^{+2}) ions (Stanish et al. 1997). The mobility of these ions would reflect how effective the pozzolan present in each mixture was at reducing the $\text{Ca}(\text{OH})_2$ available for reaction with the sulfate ions. The CNTL mixture had the highest charge passed which corroborates the highest measured expansion under sulfate attack. The mixtures with the higher 6% replacement of mS or colloidal nS also measured lower Coulomb readings, indicating reduction in the permeability and free ion availability and/or mobility through the sample. AQ6nS which had the lowest measured expansion under the 6 months of sulfate attack also had the lowest RSPT measurement of 612 Coulomb. This indicated that the nS contained mortar mixtures were more impermeable in nature and more resistant to ion transport. Likely a larger portion of the $\text{Ca}(\text{OH})_2$ was bound as secondary C-S-H due to the more active pozzolanic reactivity of the colloidal nS in comparison to mS. This higher reactivity is driven by nanosilica's higher surface area per gram, $640+ \text{ m}^2/\text{g}$ for the colloidal nS used in this study versus $22.65 \text{ m}^2/\text{g}$ for the mS. Nanosilica particles have a significantly higher surface area compared to mS due to their smaller size. The FESEM images taken of the nS and mS at similar levels of magnification visually revealed the significant difference in particle size between both forms of silica, see *Figure 4* and *Figure 5*. The presence of a more reactive pozzolanic compound in the cementitious matrix limits the availability of calcium (Ca^{+2}) and hydroxide (OH^-) ions freed by the more soluble $\text{Ca}(\text{OH})_2$ for reaction with sulfates, thus resulting in lower expansion due to reduced formation of deleterious gypsum and ettringite.

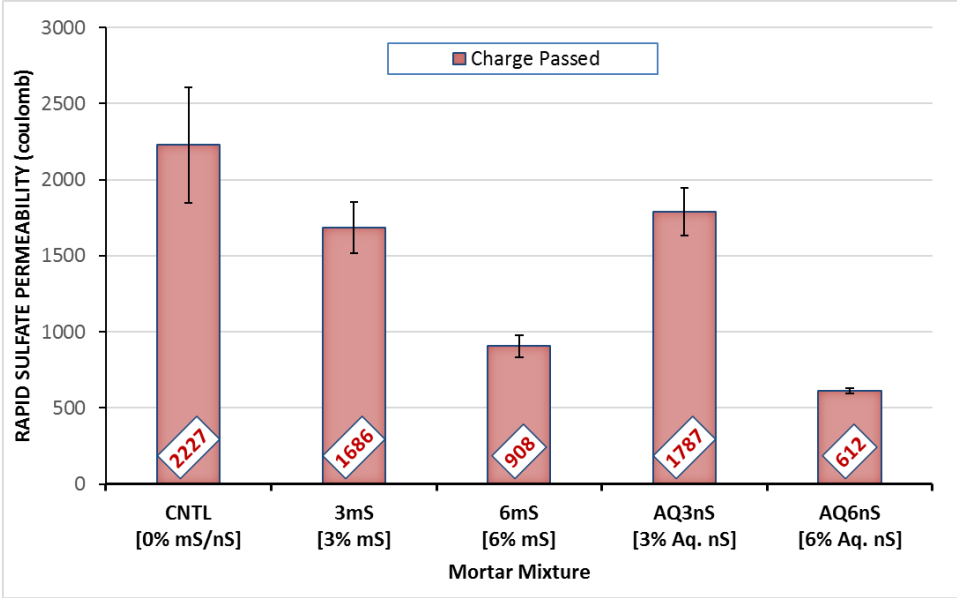


Figure 3: RSPT Results for Mortar Mixtures

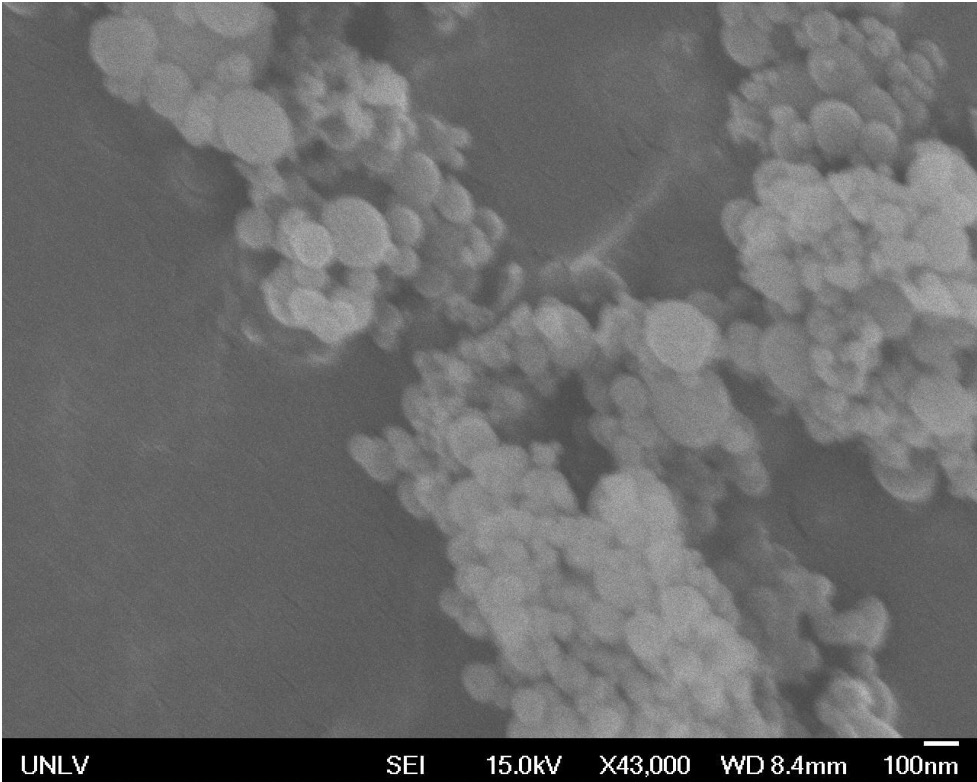


Figure 4: FESEM of Silica Fume (Microsilica)

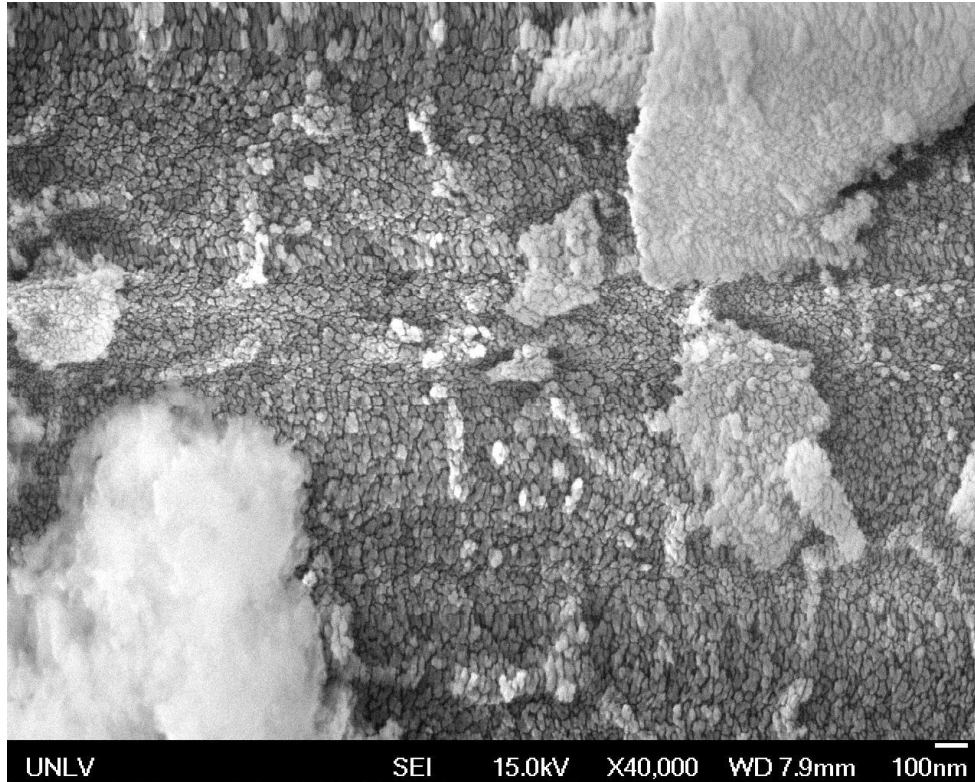


Figure 5: FESEM of Colloidal Nanosilica

The RSPT test showed a reduction in total charge passed between 3mS and 6mS but that reduction was not similarly reflected in the expansion measurements (i.e. 3mS and 6mS experienced similar expansion). The absorption test, the results of which are presented in *Figure 6*, offered an explanation. The 6mS mortar showed an increase in the total volume of permeable pore space in comparison to 3mS, 10.48% versus 9.82%. As with pure water used for the absorption test, the increased total volume of permeable voids allowed a larger quantity of sulfate solution to penetrate within the mortar sample counteracting the positive influence of mS in reducing available calcium hydroxide. In fact, the results of the absorption test showed that more sulfate solution penetrated and was available for expansive reaction in 6mS than 3mS due to its more permeable microstructure. On the other hand, the results of RSPT test proved that less calcium hydroxide was available in 6mS than 3mS as a result of a higher degree of pozzolanic reactions in 6mS than 3mS. As sulfate and calcium hydroxide are the two primary components for the expansive reactions, their effects counteracted each other resulting in a similar expansion for 3mS and 6mS. Were the pozzolanic effects of nS and mS nullified, the mixture with the higher permeable pore space would have likely exhibited more expansion than the CNTL.

The opposite effect can be observed with AQ3nS which also had similar expansion to 6mS. With AQ3nS the pozzolanic content was likely not enough which is why it had a larger coulomb reading than 6mS. In lieu of that, the mixture had a smaller volume of permeable pores of 9.20% compared to 6mS, so its pore structure was physically more impermeable to sulfate ingress and, in the end, exhibited similar expansion readings to 6mS. The absorption test also validated the better sulfate resistance of the AQ6nS mixture. The smaller permeable pore volume of the 6% colloidal nS mixture supports the physical paste densification effect observed with nanosilica

applications in cementitious mixtures from other studies (Quercia and Brouwers 2010; Senff et al. 2010; Singh et al. 2013).

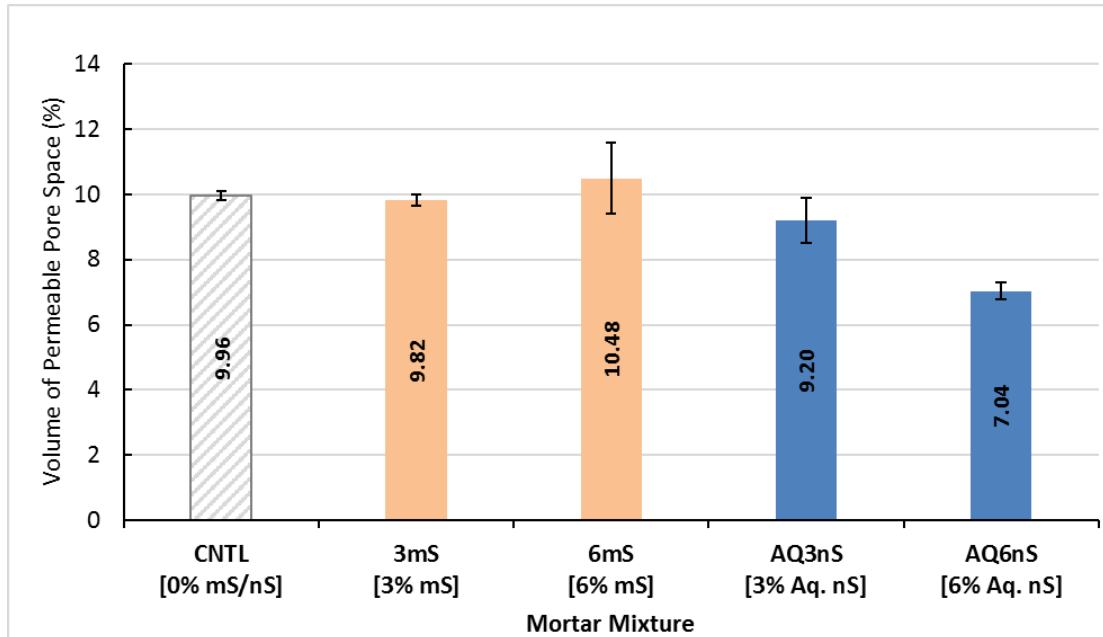


Figure 6: Absorption Results for Mortar Mixtures

Through supplemental MIP testing of the control, 6% mS and 6% colloidal nS containing mixtures, there is evidence of paste and paste-to-aggregate interfacial zone densification, as well as pore size refinement. Pore sizes in hydrated cement paste have been grouped and classified depending their effect on the strength, permeability and durability properties (Mindess et al. 2003; Neville 1998). A mortar may have a higher porosity as measured by absorption, but it may be composed of smaller more tortuous and impermeable voids or larger entrapped air voids that do not facilitate the generation of the expansive stresses that lead to volume instability and cracking from sulfate attack (Richardson 2002). Gel pores ($\leq 5\text{nm}$ [$0.005\ \mu\text{m}$]) are integral to the C-S-H phase and do not contribute to transport properties. Pores ranging from 5 to 50 nm ($0.005 - 0.050\ \mu\text{m}$) are considered capillary micropores and although tortuous, these can in small part contribute to permeability. The bulk of permeability and diffusivity occurs in the capillary macropores ranging from 50-10,000 nm (0.05 to $50\ \mu\text{m}$) that are more likely to offer pore interconnectivity (Du et al. 2014; Tobón et al. 2015).

Referring to *Figure 7*, the void volume proportion of gel pores and capillary micropores for 6mS and AQ6nS are larger in comparison to the CNTL mixture as is evident by the raised pore size distribution curves for 6mS and AQ6nS within the gel pore and micropore range. Within the capillary macropore range there is also a shift of the pore size distribution to the left towards smaller diameter pore sizes likely less conducive to permeability. Total intrusion volumes are presented in *Figure 8*, there is an overall reduction in the volume of capillary macropores for both pozzolan containing mixtures in comparison to the CNTL. Both 6mS and AQ6nS also reflect the increase on the total volume of gel pores as was noted in *Figure 7* which attests to the pozzolanic activity responsible for producing additional C-S-H phase.

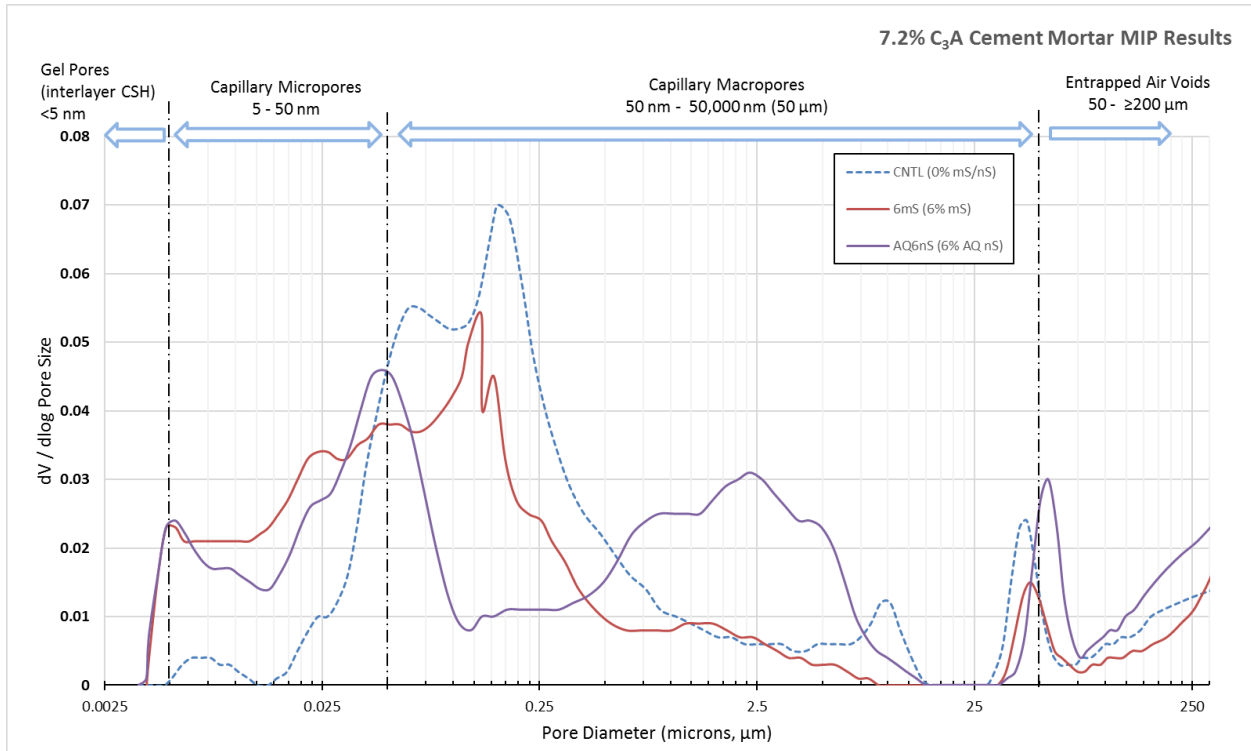


Figure 7: MIP Pore Diameter Distribution for CNTL, 6mS, and AQ6nS

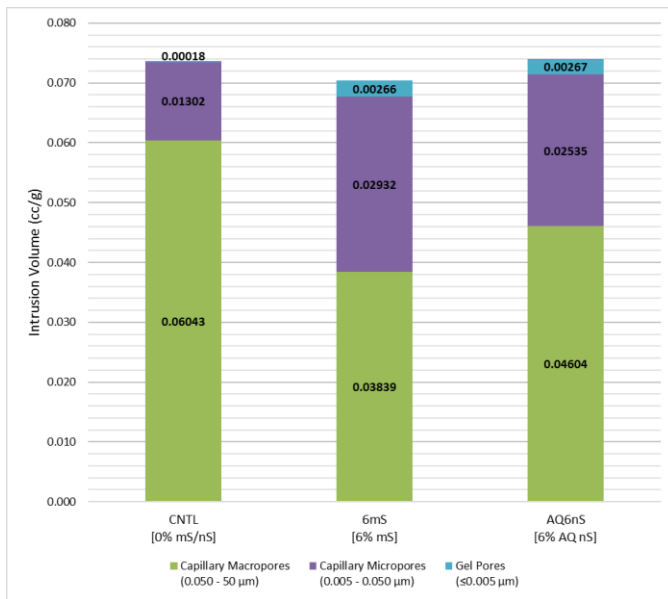


Figure 8: MIP Intrusion Volume in CNTL, 6mS, and AQ6nS

CONCLUSION

The following observations can be made from this study:

- Mortar bar sample expansion measurements over the testing period indicated that at 6% cement replacement, colloidal nS exhibited less expansion than the mS containing counterpart. Lower levels of expansion correlate to a higher resistance to external sulfate attack.
- At 3% replacement either form of silica reduced sulfate attack related expansion to a similar degree in comparison to the control. The expansion results indicate that at smaller doses mS would offer comparable performance to colloidal nS. At higher levels of replacement, the benefits of the nanoscale sized silica particles resulted in measurably superior sulfate durability performance in comparison to mS.
- The RSPT testing supports the expansion results, the nS containing mortar mixtures were more impermeable in nature and more resistant to ion transport. Likely more of the $\text{Ca}(\text{OH})_2$ was bound as secondary C-S-H due to the more active pozzolanic reactivity of the nS compared to mS.
- Absorption testing revealed that there is a larger volume of permeable pore space in the 6% mS containing mortar mixture in comparison to the 6% colloidal nS containing mixture. The smaller permeable pore volume of the 6% nS mixture supports the physical paste densification effect observed with nanosilica applications in cementitious mixtures from other studies.
- Mercury intrusion porosimetry (MIP) tests of the control and 6% silica containing mixtures, indicated evidence of paste and paste-to-aggregate interfacial zone densification, as well as pore size refinement. The void volume proportion of gel pores and capillary micropores increased in comparison to those of the control, and there was an overall reduction in the capillary macropores.

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